# Fabrication and Comparative Properties of Sustainable Epoxy Methacrylate of Bisphenol-C-Jute/Treated Jute-Natural Fibers Sandwich Composites: Part-1

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# ABSTRACT

Epoxy methacrylate of bisphenol-C-jute/treated jute and their sandwich composites of white coir, brown coir, wild almond, bamboo, betel nut, and palmyra were prepared by a compression molding technique under 5MPa pressure and at room temperature for three h. The neat sample showed almost double tensile strength than its jute composite, while it is comparable for treated jute. The composites revealed substantially improved flexural strength compared to neat. The neat, jute/treated jute and their sandwich composites indicated good impact strength, pretty good Barcol hardness, and fairly good electric strength. The neat sample showed excellent volume resistivity, while jute/treated jute and their sandwich composites revealed reasonably good volume resistivity. The composites showed high water ingress tendency, excellent hydrolytic stability, and different diffusivity in different environments at 30°C. Water ingress trend was observed the same for all the composites in different environments, and it is H<sub>2</sub>SO<sub>4</sub> > HCl >NaOH >H<sub>2</sub>O>NaCl. The sandwich composites may find their applications as low-cost housing and insulating materials and also for marine vessels.

KEYWORDS: Fiber-reinforced composites, Bio-fibers. Mechanical and electrical properties, Water ingress, Hydrolytic stability, Diffusivity.

# INTRODUCTION

Humankind uses natural fibers to make household materials such as ropes, clothes, bedsheets, packaging materials, etc. The demand for natural fibers is increasing globally day by day due to their easy and bountiful availability, biodegradability, economically cheaper than synthetic fibers<sup>[1-3]</sup>. Material scientists are looking for sustainable and nonbiodegradable fiber-reinforced composites that are accustomed to using building construction, agriculture, packaging, automobile,

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engineering, and aerospace fields to avoid environmental problems. Attractive features of natural fiber-reinforced composites are their low cost, lightweight, nonhazardous, moderate mechanical strength, and easy handling. Natural fibers can be extracted from various parts of the plants such as stems, fruits, leaves, seeds, husks, kernels, etc.<sup>[4-8]</sup> Commonly used natural fibers are jute, bamboo, coconut, ramie, kenaf, sugar cane, sisal, pineapple, abaca, nettles, rice, flax, grass, etc. The chemical compositions and water absorption tendencies of different fibers are different. [4-6] The main drawbacks of natural fibers are their high flammability [9-11] and their highly hygroscopic nature, which restrict their applications. The primary chemical constituent of natural fibers is cellulose containing a large number of hydrophilic hydroxyl groups, showing poor wettability with hydrophobic resins and poor interfacial bonding. Such drawbacks can be improved by chemical or physical modification of the fibers.<sup>[4]</sup>

The most commonly used matrices for synthetic and natural fiber-reinforced composites are epoxy, phenol-formaldehyde, and polyester resins because of their very good to excellent physicochemical properties, relatively cheaper, and readily available. The sustainability of the composites relies on many factors, namely nature and constituent fibers and matrices, pollution, temperature, environmental humidity, light, and magnetic radiations, etc.<sup>[3, 12-14]</sup> Fiberreinforced composites are used in the automotive, shipping, gas, and oil tube industries.<sup>[1, 15, 16]</sup>

Several researchers have carried out their researches on low-cost and sustainable natural fiber-reinforced composites for specific industrial applications. Most of them have used readily available commercial epoxy resin or unsaturated polyester resin based on bisphenol-A as a matrix material and achieved good to excellent physicochemical properties.

Previously we have synthesized semisolid epoxy methacrylate of bisphenol-C and prepared its jute, glass as well as biomass-filled sandwich composites<sup>[17,18]</sup> and achieved good physical properties. The semisolid resin was found difficult to handle so we have synthesized liquid epoxy methacrylate of bisphenol-C<sup>[19]</sup> and achieved comparable physical properties with commercial resin Aeropol 7105. A 40% styrene was used as a reactive diluent. Neat and 5% MMT-filled samples were prepared at room temperature by using 1% MEKP as an initiator, 1.5% dimethylaniline as a promoter, and 1% cobalt octoate as an accelerator. Mechanical and electrical properties as well as water absorption behavior of MMT filled and unfilled samples were investigated and compared. MMT filled and unfilled samples showed better studied mechanical and electrical properties and very low water ingress.

In our recent work, we have prepared 5% montmorillonite filled and unfilled epoxy methacrylate of bisphenol-C-jute, treated jute, glass, and their hybrid composites; and investigated their physicochemical properties.<sup>[20]</sup> Filled and unfilled composites revealed good to excellent mechanical and electrical properties, and excellent chemical resistance against water acids, alkali, and salt solutions. Hybrid composites showed intermediate properties of their parent composites. MMT filled treated jute (46), and unfilled glass (48), and MMT filled glass (48) composites revealed Barcol hardness between the suggested range of 45-65 for scratch and wear resistance proof materials. We had observed the effect of the nature of the

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reinforcements, matrix, filler, fiber treatment, and environmental conditions on studied mechanoelectrical properties and chemical resistance.

Using twelve different chopped (1-2 mm) biofibers, glass-bio-fibers-epoxy methacrylate of bisphenol-C sandwich composites were fabricated<sup>[21]</sup>. The sandwich composites revealed good mechanical and good to excellent electrical properties and excellent chemical resistance against water, acid, and alkali solutions. Compared to neat and glass composite the sandwich composites displayed the lowering of some of the mechanical and electrical properties because of the random orientation of the bio-fibers and poor adhesion.

We intended to protect the environment partially and support producers financially. So we

have fabricated sustainable and low-cost sandwich composites of liquid epoxy methacrylate of bisphenol-C<sup>[19]</sup> and evaluated their mechanoelectrical properties and water ingress in different environmental conditions. Such sandwich composites may replace plywood for specific applications, especially during natural calamities like floods, earthquakes, cyclones, etc. For this purpose, we had collected several natural fibers from different natural resources. Studied properties of the sandwich composites are collated with neat and glass composite.

# **EXPERIMENTAL**

## Materials and Methods

Laboratory grade solvents and chemicals used in this work were purified before their use <sup>[22]</sup> or used as received. The liquid epoxy methacrylate of bisphenol-C (EMABC) <sup>[19]</sup> was synthesized according to the following **Scheme-I**:



Scheme-I: The synthesis of epoxy methacrylate of bisphenol-C

A 40% styrene solution of EMABC was prepared at room temperature, and 0.2% hydroquinone was added as an inhibitor. This solution was designated as EMABCS, which had 1.12 g cm<sup>-3</sup> density, 300 cP viscosity, 20 min gel time, and 135°C peak exotherm temperature<sup>[19]</sup>. Methylethylketone peroxide (MEKP), 6% cobalt octoate, and mylar film were supplied by EPP Composites, Rajkot, as free samples and were used as received. Jute fabric (J) (350 GSM) was purchased from a local market (Rajkot). Jute was treated with 4% alkali solution at room temperature for 12 h<sup>[23]</sup> and designated as TJ.

Various bio-fibers (BFs) such as white coir (WC) (*Cocus Nucifera*), brown coir (BC) (*Cocus Nucifera*), were obtained from a local market (Rajkot). Wild almond fruits were procured from Junagadh and fibers (WA) (*Sterculia foetida*) were extracted manually<sup>[7]</sup>. Bamboo (BM) fibers (*Phyllostachys aurea*) were acquired from Kolkata. Betel nut (BN) (*Areca catechu*) and palmyra (PM) (*Borassus flabellifer*) fibers were bought from Calicut. Collected natural fibers were cleaned, washed several times with distilled water, and dried in an oven at 50°C for 48 h. Dry and cleaned bio-fibers were chopped into 1-2 mm size and stored in airtight containers. Chemical compositions and physical properties of some of the natural fibers are found variable<sup>[4-6]</sup> (31-83% cellulose, 18-33% hemicellulose except coir, 14-30% lignin, and 8.5-17% water absorption tendency).

## Preparation of Controls and the Composites

EMABCS control was prepared according to our recently published work<sup>[19]</sup>. The required quantities (Table 1) of EMABCS, 1% initiator (MEKP), 1% cobalt octoate (accelerator), and 1.5% dimethylaniline (promoter) of resin were transferred into a 500 mL beaker at room temperature and mixed well with a glass rod to prevent air bubble formation. Jute fabric of 20 cm x 20 cm sizes (5 plies) was impregnated by a smooth brush and kept at room temperature for 15-20 min. The chopped natural fibers were divided into four equal portions (about one-third of jute fabric) and spread uniformly on five impregnated sheets, stacked one over the other between two mylar films, and pressed between two

Composite	EMABCS, g	Jute fabric, g	Bio-fiber, g	Total wt, g
EMABCS-J	43	70	-	113
EMABCS-TJ	43	70	-	113
EMABCS-J-WC	70	72	27	169
EMABCS-TJ-WC	65	72	24	161
EMABCS-J-BC	65	70	24	159
EMABCS-TJ-BC	65	70	24	159
EMABCS-J-BN	65	70	24	159
EMABCS-TJ-BN	65	70	24	159
EMABCS-J-PM	66	72	23	161
EMABCS-TJ-PM	65	70	24	159
EMABCS-J-WA	70	72	27	169
EMABCS-TJ-WA	65	70	24	159
EMABCS-J-BM	65	70	24	159
EMABCS-TJ-BM	67	72	25	164

TABLE 1. The experimental details for the fabrication of EMABCS-J/TJ and EMABCS-J/TJ-BF sandwich composites

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mold platen under 5 MPa pressure at room temperature for three h and post cured in an oven at 150°C for 30 min; cooled to room temperature, peeled off mylar films, and machined the edges. For tensile and flexural testing, the samples were prepared as per standard test methods. For tensile strength measurements, the sample widths and thicknesses were 9.04-11.42 mm and 4.25-6.17 mm, respectively, and they were 12.03-12.49 mm and 4.25-6.27 mm, respectively for flexural strength measurements. For the water ingress study, 2cm x 2cm samples were cut from the respective sheets and sealed using corresponding resin solutions and cured at room temperature, and post-cured in an oven.

#### **Materials Characterization**

Tensile strength (ASTM-D-638-01) and flexural strength (ASTM-D-790-03) measurements were carried out at a speed of 10 mm min<sup>-1</sup> on a W & T Avery Ltd. Type 1010 Model No E-46234 (Birmingham, England). Izod impact strength (ASTM-D-256-06) tests were carried out on an Izod Impact Tester Model No.E-46204 Type A-1300 (Birmingham, England). Electric strength (IEC-60243-Pt-1-1998) quantifications were carried out in the air using 25/75 mm brass electrodes on a high voltage tester (Automatic Electric-Mumbai). Volume resistivity

(ASTM-D-257-2007) tests were carried out in the air at 25°C after 60 s charging at 500 V DC applied voltage on a Hewlett Packard high resistance meter. Replicate tests were managed 3-5 times, and average values were considered. Water ingress (ASTM-D-570-98) testing was performed at  $30\pm2°$ C by the change in mass method. The samples were weighed and dipped in water, 10% aq. NaCl, 10% aq. NaOH, 10% aq. HCl and 10% aq. H<sub>2</sub>SO<sub>4</sub> solutions at  $30\pm2°$ C. The samples were withdrawn at the interval of 24 h; the sample's surfaces were cleaned with tissue papers, reweighed, and immediately dipped in the respective solutions. The experiments were continued until the saturation/ equilibrium was established.

#### **RESULTS AND DISCUSSION**

## **Mechanical Properties**

Comparative mechanical properties namely tensile strength, flexural strength, flexural modulus, Izod impact strength, and Barcol hardness of EMABCS (neat), EMABCS-J/TJ, and EMABCS-J/TJ-BF are shown respectively in Figures1-5. The tensile strength of the neat



Figure 1. Comparative tensile strength of EMABCS-J/TJ and EMABCS-J/TJ-BF.



Figure 2. Comparative flexural strength of EMABCS-J/TJ and EMABCS-J/TJ-BF.



Figure 3. Comparative flexural modulus of EMABCS-J/TJ and EMABCS-J/TJ-BF.

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Figure 4. Comparative impact strength of EMABCS-J/TJ and EMABCS-J/TJ-BF.



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sample is almost double that of EMABCS-J and is comparable with EMABCS-TJ. The enhanced tensile strength of the alkali-treated composites is due to surface modification of jute fibers and improved interface adhesion. EMABCS-TJ-BF resulted in enhanced tensile strength than EMABCS-J-BF due to the abovementioned reasons. EMABCS-J-WC/BC/BN/ WA/BM showed good to excellent better tensile strength (23.1-92.3%). EMABCS-J-PM showed comparable tensile strength. Similarly, EMABCS-TJ-WC/BC/BN/WA ensured 12.1-34.5% good tensile strength. Thus, both types of the sandwich composites showed improved tensile strength relative to EMABCS-J and EMABCS-TJ except EMABCS-J-PM, EMABCS-TJ-PM, EMABCS-J-BM, and EMABCS-TJ-BM. Thus, different constitutional compositions and physical properties of the reinforcements<sup>[4-6]</sup>, surface modification of the jute fibers, the nature of the matrix, and interface adhesion affected the tensile strength of the composites.

From Figures 1 and 2, it is clear that the flexural strength of the neat sample is somewhat smaller than the tensile strength indicating the rigid nature of the neat sample. The flexural strength of the treated and untreated jute composites is enhanced substantially over EMABCS because of the stiff nature of the jute fibers. EMABCS-TJ showed almost doubled flexural strength than EMABCS-J and EMABCS (2.6 times). The enhancement of flexural property is mainly due to surface modification and improved interface bonding. Interface adhesion can be better judged by SEM analysis but due to personal limitations, such analysis was not done. We had judged the enhancement or deterioration of the

properties based on observed results. The stated sandwich composites revealed a significantly improved flexural strength in light of the neat sample. EMABCS-J-WC/BN/PM/ WA/BM displayed 15-40% more flexural strength except for EMABCS-J-BC due to the inherent characteristic properties of the biofibers. EMABCS-TJ-WC demonstrated slightly better flexural strength (8%) than EMABCS-TJ, and EMABCS-TJ/BC/BN/PM/WA/BM indicated slightly lower flexural strength (2-6%). The lowering of the tensile and flexural properties of EMABCS-TJ-BF sandwich composites is probably due to the increased polarity of treated jute fibers, which increased like interactions of hydroxyl groups of jute and bio-fibers with matrix and thereby lowering interface adhesion. The mechanical properties of the bio-fiber reinforced composites are governed by the selection of the bio-fibers, and the matrix, the chemical compositions, microfibril angles, fiber surface treatment, fabrication technique, fiber orientation, fiber strength, and stiffness, fiber loading, etc.<sup>[24-26]</sup>. In the present work, we had used bioriented jute fabric and randomly oriented bio-fibers in making composites. Untreated bio-fibers have noncellulosic compounds and smooth surfaces, which lead to inferior mechanical interlocking and incompatibility between bio-fibers and polar matrix material. Fiber modification can improve mechanical properties because of improved compatibility, hydrophobicity, interfacial bonding, and surface roughness of bio-fibers.

Figure 3 revealed excellent flexural modulus of EMABCS-J/TJ, and EMABCS-J/TJ-BF due to improved stiffness of the composites. The observed fact is due to reinforcement of the bioriented jute fibers, randomly oriented bio-

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fibers, and fiber treatment. The flexural moduli of EMABCS-J and EMABCS-J-BF are much smaller than EMABCS-J and EMABCS-TJ-BF. EMABCS-TJ has indicated 1.4 times more stiffness than EMABCS-J. EMABCS-J/TJ is 1.2-1.3 and 1.1-1.7 times more rigid than EMABCS-J/TJ-BF. Somewhat decrease in stiffness of the sandwich composites is mainly due to the random orientation of bio-fibers and their characteristic properties. In sandwich composites, stress transfer is discontinuous, which causes a lowering in the mechanical properties of the composites. In bioriented jute/ treated jute fibers, stress transfer is continuous, and as a result, mechanical strengths are improved. The overall result of the use of oriented and randomly oriented fibers is the deterioration of the properties of the composite materials.

EMABCS, EMABCS-J/TJ, and EMABCS-J/TJ-BF showed good impact strength (Figure 4). EMABCS-J (17%) and EMABCS-TJ (43.5%) showed better impact strength than EMABCS due to improved interfacial bonding between jute and matrix material. Similarly, EMABCS-J (18-59 %) and EMABCS-TJ (21-42 %) displayed better impact strength than EMABCS-J/TJ-BF. The random orientation of the bio-fibers in the sandwich composites is mainly responsible for the decreasing impact strength.

It is noticed from Figure 5 that the mentioned composites showed enhanced Barcol hardness. EMABCS and EMABCS-J have equivalent Barcol hardness and it is slightly better for EMABCS-TJ due to alkali treatment. EMABCS-J-BF (19-36%) and EMABCS-TJ-BF (5-30%) showed reduced Barcol harness than EMABCS-J and EMABCS-TJ. The decrease in Barcol hardness of the sandwich composites is mainly due to the random orientation and variable chemical and physical properties of the bio-fibers. The fiber modification followed almost identical Barcol hardness of EMABCS-TJ-WC/BC/BN/BM and EMABCS. None of the samples can be used as scratch and wear resistance materials because of the lack of the required Barcol hardness of 45-65<sup>[27]</sup>. The acquired mechanical properties of the studied composites suggested their utility as low loadbearing housing applications.

# **Electrical Properties**

The electric strength and volume resistivity of neat, untreated jute, treated jute, and sandwich composites are represented in Figures 6 and 7, respectively. The electric strength of the neat and EMABCS-TJ is comparable and almost double that of EMABCS-J (Figure 6). Both types of the sandwich composites revealed good electric strength except EMABCS-TJ-WC/BC/ WA. EMABCS-J-WC/BC/BN/PM/WA (16-63%) showed better electric strength than EMABCS-J and EMABCS-J-BM displayed equivalent strength. The electric strength of EMABCS-TJ-BN/PM/BM is declined 7-29% over EMABCS-TJ. In the case of EMABCS-TJ/WC/ BC/WA, the decline in the electric strength is substantially very high (47-76%). The declining electric strength of the treated jute sandwich composites might be due to the increased surface polarity of treated jute fibers.

The neat sample revealed excellent volume resistivity, while jute, treated jute, and sandwich composites showed fairly good to good volume resistivity (Figure 7) due to the polar nature of reinforcing fibers and their different chemical compositions. Surface modification of jute



Figure 6. Comparative electric strength of EMABCS-J/TJ and EMABCS-J/TJ-BF.





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fibers leads to increased volume resistivity of EMABCS-TJ over EMABCS-J (two times). EMABCS-TJ-WC and EMABCS-J-BC exhibited comparable volume resistivity with EMABCS-J.

Polymeric materials are dielectric materials and show electrical conductivity (inverse of resistivity) of the order of 10<sup>-12</sup> to 10<sup>-15</sup> ohm-cm due to the presence of free charge carriers <sup>[28, 29]</sup>. The observed electrical properties of the composites are mainly due to the polar nature of reinforcements and matrix. The pretty good electric strength and volume resistivity of the composites indicated their usefulness as electrically insulating materials.

## Water Ingress Study

Under environmental conditions, polymeric fiberreinforced composites undergo different interactions such as a chemical reaction, absorption, solvation, stress cracking, plasticization, etc. Consequently, their physicochemical properties change with time. Overall chemical resistance is determined based on some characteristic properties like bond strength, polarity, degree of branching, % crystallinity, temperature, etc. In a humid atmosphere, the moisture ingress in polymeric composites relies on the nature of the constituent components. It is a well-known fact that the natural fibers possess variable chemical compositions, physical properties, and water uptake (9-16%) tendencies.<sup>[4]</sup> Bio-fibers are hygroscopic because of the presence of many hydrophilic hydroxyl groups, which leads to poor wettability with hydrophobic matrix<sup>[4, 5, 23, 30, 31]</sup>.

Water ingress in EMABCS-J/TJ and EMABCS-J/TJ-BF composites was studied by assuming unidimensional Fickian type diffusivity<sup>[31]</sup>. Following the kinetics of the mass change method at room temperature  $(30\pm2^{\circ}C)$  and the interval of 24h was followed in different electrolytic solutions such as water, 10% aq. NaCl, 10% aq. NaOH, 10% aq. HCl and 10% aq.H<sub>2</sub>SO<sub>4</sub>. The percent water ingress in the composites was calculated according to Eqn.1:

$$M = rac{W_m - W_d}{W_d} x 100 \dots 1$$

Where M = % water ingress,  $W_m =$  weight of the moist sample, and  $W_d =$  weight of the dry piece.

Typical % Weight gain against Time curves for EMABCS-J/TJ-PM and EMABCS-J/TJ-WA are shown in Figures 8 and 9, respectively. Experimentally it was observed that the % weight gain was found to increase over time and then remained practically constant, i.e., equilibrium or saturation point. The saturation time and saturation water content in different environments for all the composites are reported in Table 2. The observed water ingress trend in the composites is the same in different environments. Obtained water ingress trend is H<sub>2</sub>SO<sub>4</sub> >HCl >NaOH >H<sub>2</sub>O >NaCl. Water ingress tendency in treated jute composites is reasonably lower than jute composites. The observed fact is due to surface modification. In different media (Table 2), untreated jute (20.5-17.8%) and jute-biofiber (20.5-14.1%) composites showed higher saturation water content than treated jute (16.9-14.4%) and treated jute-biofiber (17.7-14.1%). The high water ingress tendency of the composites is due to the hydrophilic nature of the jute and bio-fibers. The variable chemical compositions of the bio-fibers affected water ingress in the

composites. The stated composites showed a relatively longer saturation time (384-456h) and

excellent hydrolytic stability in harsh environmental conditions.



Figure 8. Comparative water ingress curves of EMABCS-J/TJ-PM sandwich composites at 30°C in different conditions.



Figure 9. Comparative water ingress curves of EMABCS-J/TJ-WA sandwich composites.

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Hemicellulose leads to higher water ingress in the composites. In contrast, lignin-rich fiber composites show better resistance to weathering <sup>[32]</sup> due to a lower affinity towards moisture and act as a protective barrier for cellulose microfibrils from water ingress. The nature of electrolytes with the same concentration and different chemical compositions of bio-fibers affected saturation time and equilibrium water absorption behavior.

# Diffusivity

The diffusivity in EMABCS-J/TJ and EMABCS-J/TJ-BF in a given environment was evaluated from the initial slopes of M against  $\sqrt{t}$  curves according to Eqns. 2 and 3:

$$M = \frac{4M_{\rm m}}{\rm h} \sqrt{\frac{\rm t}{\pi}} \sqrt{\rm D_x} \dots 2$$
$$D_{\rm x} = \pi \left(\frac{h}{4M_{\rm m}}\right)^2 (slope)^2 \dots 3$$

TABLE 2:	Saturation	time a	and	saturation	water	content	data	of	EMABCS-J/TJ	and	EMABCS-J/TJ-BF	sandwich
composite	es.											

Composite	Equilibrium water content, %					Equilibrium time, h				
	H <sub>2</sub> O	10% NaCl	10% NaOH	10% HCI	10% H <sub>2</sub> SO <sub>4</sub>	H₂O	10% NaCl	10% NaOH	10% HCI	10% H <sub>2</sub> SO <sub>4</sub>
EMABCS-J	18.8	17.8	19.6	20.1	20.5	408	384	408	384	408
EMABCS-TJ	15.2	14.4	15.6	16.5	16.9	432	432	432	432	432
EMABCS-J-WC	19.0	14.6	19.5	19.9	20.2	432	432	432	432	408
EMABCS-TJ-WC	16.1	15.3	16.5	16.7	17.7	432	432	432	408	432
EMABCS-J-BC	18.9	17.7	19.6	20.0	20.4	432	432	432	456	432
EMABCS-TJ-BC	16.0	15.1	16.5	17.3	17.7	432	408	432	456	432
EMABCS-J-BN	18.4	17.9	19.4	20.2	20.5	408	432	408	432	432
EMABCS-TJ-BN	15.8	15.1	16.1	17.1	17.6	432	432	408	432	408
EMABCS-J-PM	19.6	17.9	18.3	19.4	20.3	408	432	456	456	456
EMABCS-TJ-PM	15.1	14.1	15.6	16.4	16.9	432	432	432	432	432
EMABCS-J-WA	19.	18.3	19.2	20.0	20.4	432	432	456	432	408
EMABCS-TJ-WA	16.1	14.6	15.9	16.5	17.3	408	432	384	408	432
EMABCS-J-BM	18.6	17.5	19.4	20.6	20.8	432	432	432	432	432
EMABCS-TJ-BM	15.4	14.3	16.1	16.3	17.1	432	408	384	384	408

Where Dx = diffusivity, t = time (second),  $h = sample thickness (m) and <math>M_m = equilibrium$  water ingress. Evaluated diffusivities in the

composites in different environments are presented in Table 3. The environments, type, and nature of the constituents of the

composites and strong electrolytes affected the diffusivity in the composites. The different electrolytes affect water structure differently, and as a result, different diffusivity is observed in different environments due to different solvated ionic sizes [17-19, 23].

Water diffusion in the composites occurred through the capillary mechanism and resulted in swelling, and blistering. The diffusion in the composites was continued with time till saturation point and then free water was occupied in the voids if any. Due to personal limitations, the void content in the composites was not carried out. Free water may interact with constituent components of the composite and thereby may cause delamination, degradation, or void formation<sup>[33]</sup>. Absorbed moisture causes swelling and plasticization of resin, weakening of interface increases delamination rate and thereby deterioration of the mechanical properties of the composite.<sup>[13, 33-39]</sup> Cracking and blistering lead to high water absorption, while the leaching of small molecules results in a decrease in weight <sup>[40]</sup>. In the present case, no leaching from the composites was observed. Excellent hydrolytic stability and longer equilibrium time of the sandwich composites in different media indicated their usefulness in harsh and humid environments.

## CONCLUSIONS

Epoxy methacrylate of bisphenol-C-jute/treated and their biofiber sandwich composites was

Composite	Diffusivity (Dx), 10 <sup>-13</sup> , m <sup>2</sup> s <sup>-1</sup>									
	H <sub>2</sub> O	10% NaCl	10% NaOH	10% HCI	10% H₂SO₄					
EMABCS-J	1.9	2.0	1.8	1.7	2.1					
EMABCS-TJ	1.8	2.3	2.4	1.9	2.1					
EMABCS-J-WC	1.2	1.5	1.4	1.2	1.1					
EMABCS-TJ-WC	1.1	1.3	1.5	1.9	1.3					
EMABCS-J-BC	1.7	2.3	2.1	1.5	1.7					
EMABCS-TJ-BC	1.2	2.1	1.8	1.5	1.7					
EMABCS-J-BN	1.7	1.4	2.2	1.3	1.8					
EMABCS-TJ-BN	1.8	1.4	1.7	2.6	2.2					
EMABCS-J-PM	1.3	1.4	1.1	2.0	2.1					
EMABCS-TJ-PM	1.9	1.4	1.2	1.5	1.9					
EMABCS-J-WA	1.4	1.9	1.4	1.7	2.0					
EMABCS-TJ-WA	2.2	2.3	1.9	1.6	1.9					
EMABCS-J-BM	2.2	2.1	1.5	1.8	1.5					
EMABCS-TJ-BM	3.7	2.9	2.2	1.9	2.9					

TABLE 3: Diffusivity data of EMABCS-J/TJ and EMABCS-J/TJ-BF sandwich Composites

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prepared and determined their mechanical and electrical properties; and water ingress in different media at 30°C. The composites showed good mechanical and electrical properties, and excellent hydrolytic stability, and a high water ingress tendency in various media. Treated jute and its sandwich composites revealed improved studied properties. The nature of the matrix, reinforcements, electrolytes, fiber orientation, surface modification, chemical compositions of fibers, and their physical properties affected the investigated physical properties of the composites.

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